

## Note

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### The enthalpy of formation of aluminium titanate

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(Received 4 May 1973)

The enthalpy of formation of  $\text{Al}_2\text{TiO}_5$  from the component oxides near the transition temperature has been reported to be  $272 \text{ kJ/mole}^1$ . However, during a DTA investigation of phase transformations which occur during the heat treatment of  $\text{Al}_2\text{O}_3\text{-TiO}_2$  powders condensed from a plasma, the enthalpy of formation from the oxides has been found to be an order of magnitude lower than the previously reported value.

The  $\text{Al}_2\text{O}_3\text{-TiO}_2$  powder used was prepared by the oxidation of mixed aluminium and titanium halides in an oxygen high frequency plasma torch and consisted of spherical particles approximately  $0.1 \mu\text{m}$  diameter. In the composition range 14–79 weight %  $\text{TiO}_2$ , each particle consisted of a finely dispersed two phase mixture of  $\delta\text{-Al}_2\text{O}_3$  and rutile<sup>2</sup>.

Differential thermal analysis of the powders was carried out using a Rigaku model M8076 micro-DTA TGA apparatus at a heating rate of  $20^\circ\text{C min}^{-1}$  to a maximum temperature of  $1500^\circ\text{C}$  with  $\alpha\text{-Al}_2\text{O}_3$  as reference material. An exothermic peak, the temperature of which varied between 1000 and  $1240^\circ\text{C}$  depending upon the  $\text{TiO}_2$  content, and an endothermic peak which began at 1340 and was complete at  $1375^\circ\text{C}$ , independent of  $\text{TiO}_2$  content, were observed. X-ray diffraction of heat treated powders showed that the exotherm was due to the transformation of  $\delta\text{-Al}_2\text{O}_3$  to  $\alpha\text{-Al}_2\text{O}_3$ , and the endotherm to the formation of  $\text{Al}_2\text{TiO}_5$  from  $\alpha\text{-Al}_2\text{O}_3$  and rutile. The enthalpies of the transformations were determined from the areas under the DTA peaks using the known enthalpy of transition of  $5.94 \text{ kJ mole}^{-1}$  at  $1200^\circ\text{C}$  for the tetragonal to monoclinic phase change in zirconia<sup>3</sup> for calibration. Zirconia was chosen because of the similarity of the temperature and enthalpy of the transformation to those for the transformations of  $\gamma\text{-Al}_2\text{O}_3$  and  $\delta\text{-Al}_2\text{O}_3$  to  $\alpha\text{-Al}_2\text{O}_3$ ; "Ugine Kuhlmann" 99.9% zirconia powder was used. The enthalpies of transformation of pure  $\delta\text{-Al}_2\text{O}_3$  powder prepared by the plasma method and "Linde B"  $\gamma\text{-Al}_2\text{O}_3$  powder to  $\alpha\text{-Al}_2\text{O}_3$  were also determined. The mean value of seven measurements for  $\delta\text{-Al}_2\text{O}_3$  was  $-11.7 \text{ kJ mole}^{-1}$  with a range of  $-9.6$  to  $-13.8$  and the mean value of four measurements for  $\gamma\text{-Al}_2\text{O}_3$  was  $-24.3 \text{ kJ mole}^{-1}$  with a range of  $-18.8$  to  $-31.4$ . Several values for the enthalpy of transition from  $\gamma\text{-Al}_2\text{O}_3$  to  $\alpha\text{-Al}_2\text{O}_3$  have been reported with  $\Delta H^\circ$  ( $t = 25^\circ\text{C}$ ) ranging from  $-18.4$  to  $-63.2 \text{ kJ mole}^{-1}$ <sup>4,5</sup>; the present results for both  $\gamma\text{-Al}_2\text{O}_3$  and  $\delta\text{-Al}_2\text{O}_3$  however agree reasona-

bly well with the values derived from enthalpies of solution of  $-22.2$  and  $-11.3$  kJ mole $^{-1}$  at  $750^{\circ}\text{C}$ <sup>6</sup>.

A mean value of  $25.1$  kJ mole $^{-1}$  from seven measurements, with a range of  $22.6$ – $26.8$  was obtained for the heat of reaction of  $\alpha\text{-Al}_2\text{O}_3$  and  $\text{TiO}_2$  to form  $\text{Al}_2\text{TiO}_5$  at  $1340^{\circ}\text{C}$ . Although similar methods for the estimation of heats of reaction from DTA measurements have been reported<sup>5,7,8</sup> it is recognized that the method may be subject to relatively large experimental errors, due mainly to changes in thermal conductivity of the sample while the reaction is taking place, particle size effects, and variation in sample packing<sup>9</sup>. Kostomaroff and Rey<sup>5</sup> have shown that the thermal conductivity of fine powders is low, and changes during reaction have a negligible effect. The other effects were minimised in the present work by the small sample size ( $20$ – $30$  mg) and uniform packing of the powders using a hand press, leading to an estimated error of  $\pm 8$  kJ mole $^{-1}$  in the enthalpy of reaction.

There is evidence that the reaction  $\text{Al}_2\text{O}_3 + \text{TiO}_2 \rightarrow \text{Al}_2\text{TiO}_5$  is reversible, and equilibrium occurs at approximately  $1300^{\circ}\text{C}$ . Lang et al.<sup>10</sup> have found that  $\text{Al}_2\text{TiO}_5$  is unstable at temperatures in the range  $750$ – $1300^{\circ}\text{C}$  and stable from  $1300^{\circ}\text{C}$  to fusion. Hamelin<sup>11</sup> reported no reaction between the oxides at  $1250^{\circ}\text{C}$  and complete reaction at  $1450^{\circ}\text{C}$  and DTA measurements have given  $1380^{\circ}\text{C}$  as the reaction temperature<sup>12</sup>. At equilibrium the change in free energy is zero and the enthalpy of reaction may be estimated from the standard entropies of  $\text{Al}_2\text{TiO}_5$ <sup>13</sup>,  $\alpha\text{-Al}_2\text{O}_3$  and rutile<sup>2</sup>. Assuming equilibrium at  $1300^{\circ}\text{C}$ , the calculated value is  $\Delta H = 17.6$  kJ mole $^{-1}$ , which is reasonably consistent with the DTA results.

The temperature correction for the heat of reaction from  $1340$  to  $25^{\circ}\text{C}$ , based on heat capacity data<sup>3,4</sup> is  $1.7$  kJ mole $^{-1}$  and may be neglected. The standard enthalpy of formation of aluminium titanate from its constituent elements is then

$$\Delta H_{\text{Al}_2\text{TiO}_5}^{\circ} = H_{\text{Rutile}}^{\circ} + H_{\alpha\text{-Al}_2\text{O}_3}^{\circ} + H_{\text{React}}^{\circ} = -2.600 \text{ kJ mole}^{-1}$$

using the reported enthalpies of formation of  $\alpha\text{-Al}_2\text{O}_3$  and rutile<sup>4</sup>

This work has been supported by an Australian Research Grants Committee grant and M. S. J. Gani holds a Commonwealth Post-Graduate Award. Thanks are due to the Department of Chemical Engineering, Monash University, for the use of the DTA apparatus.

- 1 R. A. Slepetys and P. A. Vaughan, *J. Phys. Chem.*, 73 (1969) 2157.
- 2 M. S. J. Gani and R. McPherson, *J. Aust. Ceram. Soc.*, 8 (1972) 65.
- 3 E. G. King and J. P. Coughlin, *J. Amer. Chem. Soc.*, 72 (1950) 2262.
- 4 *JANAF Thermochemical Tables*, Dow Chemical Co., Mich., 1967.
- 5 V. Kostomaroff and M. Rey, *Silicates Ind.*, 28 (1963) 9.
- 6 T. Yokokawa and O. J. Kleppa, *J. Phys. Chem.*, 68 (1964) 3246.
- 7 C. N. R. Rao and K. J. Rao, *J. Mat. Sci.*, 1 (1966) 238.
- 8 B. R. Currell, in R. F. Schwenker Jr. and P. D. Garn (Eds.), *Thermal Analysis*, Academic Press, New York, 1969, p. 1185.
- 9 R. C. MacKenzie and B. D. Mitchell, in R. C. MacKenzie (Ed.), *Differential Thermal Analysis*, Vol. 1, Academic Press, New York, 1970, p. 108.
- 10 S. M. Lang, C. L. Fillmore and L. H. Maxwell, *J. Res. Nat. Bur. Stand*, 48 (1952) 298.
- 11 M. Hamelin, *Bull. Soc. Chim. Fr.*, (1957) 1421.
- 12 B. N. Battacharyya and Sudhir Sen, *Glass Ceram. Bull. (India)*, 12 (1965) 92.
- 13 K. R. Bonnickson, *J. Amer. Chem. Soc.*, 77 (1955) 2152.